

## Article

# ZnS/SiO<sub>2</sub> Passivation Layer for High-Performance of TiO<sub>2</sub>/CuInS<sub>2</sub> Quantum Dot Sensitized Solar Cells

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**Abstract:** Suppressing the charge recombination at the interface of photoanode/electrolyte is the crucial way to enhance the photovoltaic performance of quantum dot sensitized solar cells (QDSSCs). In this scenario, ZnS/SiO<sub>2</sub> blocking layer was deposited on TiO<sub>2</sub>/CuInS<sub>2</sub> QDs to inhibit the charge recombination at photoanode/electrolyte interface. As a result, the TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> based QDSSCs delivers a power conversion efficiency ( $\eta$ ) value of 4.63%, which is significantly higher than the 2.15% and 3.23% observed for QDSSCs with a TiO<sub>2</sub>/CuInS<sub>2</sub> device and TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS, respectively. Electrochemical impedance spectroscopy and open circuit voltage decay analyses indicate that ZnS/SiO<sub>2</sub> passivation layer on TiO<sub>2</sub>/CuInS<sub>2</sub> suppress the charge recombination at the photoanode/electrolyte interface and prolongs the electron lifetime.

**Keywords:** QDSSCs; Charge recombination; ZnS/SiO<sub>2</sub>; Passivation layer;

## 1. Introduction

Semiconductor quantum dots (QDs) based on II-VI group such as CdSe,[1] CdTe,[2] CdS,[3] PbS,[4] PbSe,[5] and etc. have been extensively studied for quantum dot sensitized solar cell (QDSSC) and photocatalysis applications, due to tunable band gap, high extinction coefficient, large intrinsic dipolar moment and multiple excitation generation (MEG).[6,7] However, highly toxic Cd or Pb-containing QDs based solar cells show the excellent photostability and high power conversion efficiencies (PCEs), but the intrinsic high toxicity of Cd or Pb still limit the commercial applications in consideration of environmental and health concerns. Therefore, the development high quality “green” QDs sensitizers without carcinogenic heavy metal element is critical for the practical applications of QD based solar cells.

As an alternatives to the highly toxic Cd and Pb chalcogenide QDs, less-toxic I–III–VI<sub>2</sub> group QDs, especially CuInS<sub>2</sub> (CIS) QD has been attracted due to high absorption coefficient ( $\sim 10^5$  cm<sup>-1</sup>) and near optimal band gap energy (1.0–1.5 eV), both of which make it a promising candidate as a sensitizer in QDSSCs [8–10]. There are two common approaches have been demonstrated for assembling CuInS<sub>2</sub> QDs onto TiO<sub>2</sub> electrodes: The first method uses presynthesized QDs onto TiO<sub>2</sub> surfaces through direct adsorption or bifunctional-linker-assisted adsorption and (2) the in situ preparation of QDs onto TiO<sub>2</sub> by successive ionic layer adsorption and reaction (SILAR) providing high surface coverage of QDs [11,12]. Owing to its facile and reproducible preparation, The SILAR process has recently emerged as the best method for adsorbing QDs onto TiO<sub>2</sub> electrodes with high

QD loading and well controllable in size and density of the target semiconductor QDs and efficient electron transfer to TiO<sub>2</sub> [13,14].

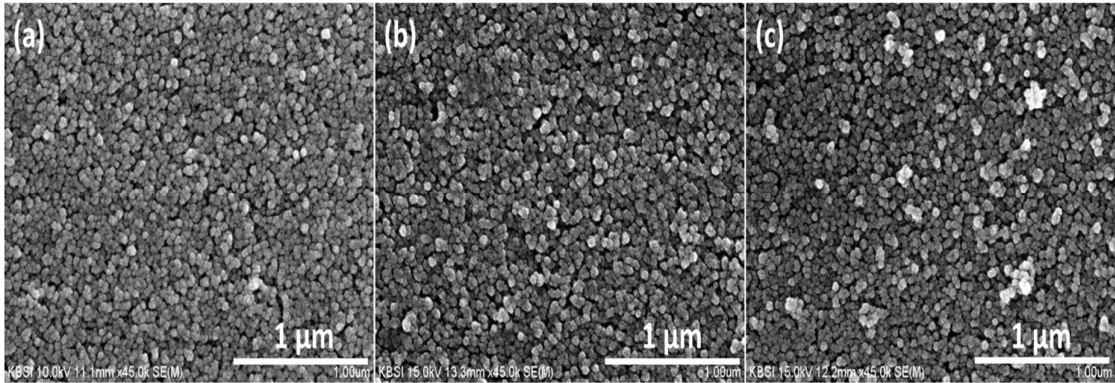
Chang *et al.* reported that CuInS<sub>2</sub> QD-sensitized TiO<sub>2</sub> nanoparticle film with a buffer layer (Cu<sub>2</sub>S) and a passivation layer (ZnSe), fabricated by using the successive ionic-layer adsorption and reaction (SILAR) process and achieved a PCE of 2.52% [15]. Zhou *et al.* reported QDSSCs based on CuInS<sub>2</sub> and introduction of In<sub>2</sub>S<sub>3</sub> buffer layer using SILAR process, which presented as high as ~1.06% PCEs [16]. However, the lower photovoltaic performance of QDSSC is due to severe charge recombination process at the interface of the TiO<sub>2</sub>/QD/electrolyte interface. To suppress charge recombination in sensitized cells, over coating the mesoporous oxide electrode with a thin wide band gap inorganic barrier layer, which act as an energy barrier preventing electrons from recombining. Until now, ZnS is the most popular photoanode coating material for preventing interfacial recombination in QDSSCs [17]. The ZnS over layer is introduced by the facile SILAR process, which covers the bare mesoporous TiO<sub>2</sub> alongside the surface of QDs toward the electrolyte. The ZnS coating was found to enhance the performance of solar cells due to the passivation of the QD surface states, resulting in suppression of the recombination processes within the cell. Interestingly, the combination of ZnS coating with different dipolar molecules absorbed on the TiO<sub>2</sub> surface has been demonstrated to improve the performance of QDSSCs [18].

Herein, we report that a novel ZnS/SiO<sub>2</sub> double barrier coating treatment, sequentially deposited after CuInS<sub>2</sub> QD on photoanode, attenuates substantially the recombination in QDSSCs. SiO<sub>2</sub> layer presents a high porosity with high micropore volume, which allows a good photogenerated hole capture by electrolyte from QD when it covers ZnS (ZnS/SiO<sub>2</sub>). As a result the ZnS/SiO<sub>2</sub> on CuInS<sub>2</sub> favors the enhancement of photocurrent, photovoltage, and consequently the PCE of the resultant cell devices. The QDSSC based on the CuInS<sub>2</sub> QD sensitizer and ZnS/SiO<sub>2</sub> double layer exhibits a PCE of 4.63% (with short circuit current density ( $J_{sc}$ ) = 12.83 mA cm<sup>-2</sup>, an open circuit voltage ( $V_{oc}$ ) = 0.603 V, fill factor (FF) = 0.598) under AM 1.5 G one full sun illumination, which is much higher than the CuInS<sub>2</sub>/ZnS (PCE = 3.23%) and CuInS<sub>2</sub> (PCE = 2.15 %). Electrochemical impedance spectroscopy (EIS) and open circuit voltage decay (OCVD) measurements show that this ZnS/SiO<sub>2</sub> double layer significantly reduces the electron recombination at the photoanode/electrolyte interfaces.

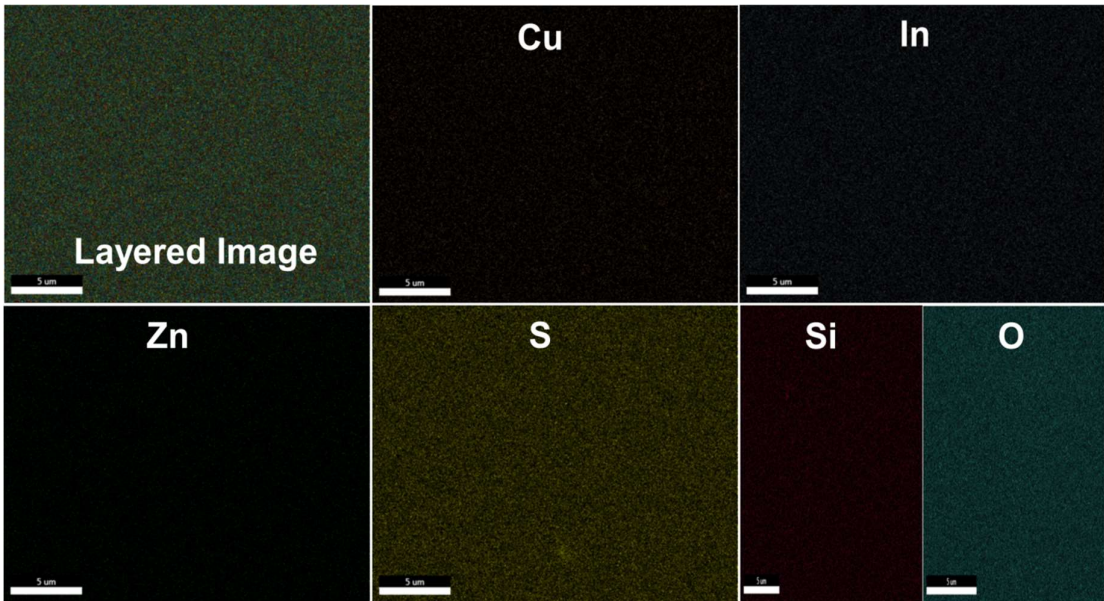
## 2. Results and Discussion

### 2.1. Morphological characterization

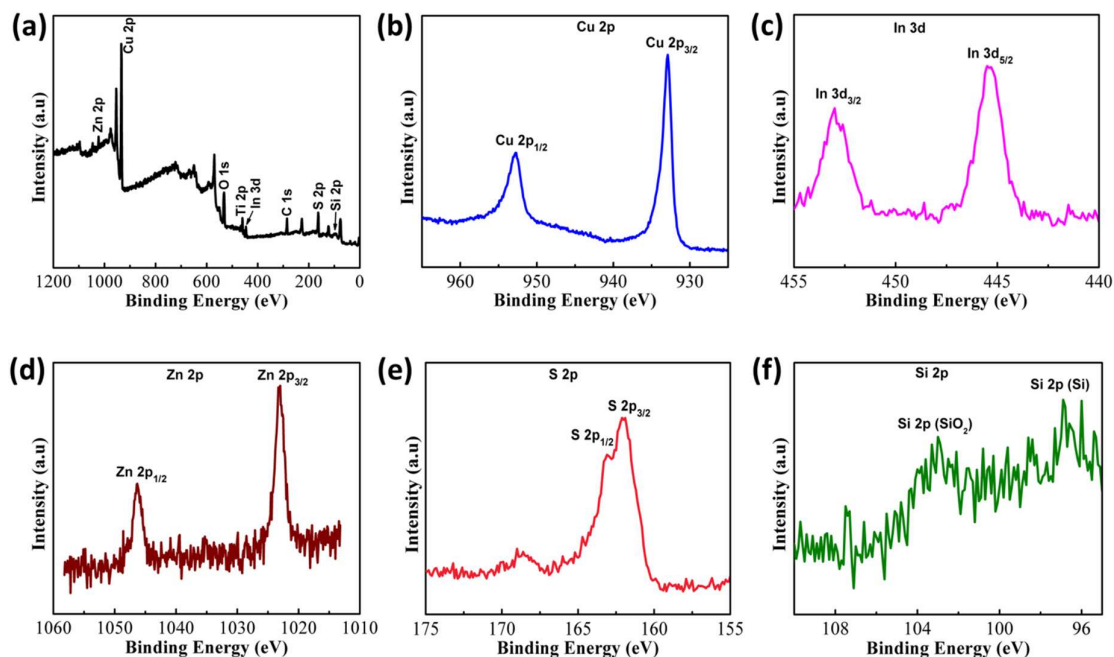
Figure 1a, 1b, and 1c show the SEM images of the bare CuInS<sub>2</sub>, CuInS<sub>2</sub>/ZnS, and CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> layers on the surface of TiO<sub>2</sub>. The bare CuInS<sub>2</sub> film in Figure 1b exhibits uniformly distributed nanoparticles on the TiO<sub>2</sub> surface. The surface morphology of the samples were similar although there is a slight increase in the particle size of with the ZnS and ZnS/SiO<sub>2</sub> layer deposition; however, the exact change in size of particles is difficult to investigate and was inconclusive. Therefore, EDX and XPS measurements were conducted to identify the ZnS and ZnS/SiO<sub>2</sub> coatings on the surface of TiO<sub>2</sub>. Moreover, the compositional distributions of a TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> sample are further demonstrated by elemental mapping behavior, in which the homogeneous distribution and coexistence of Cu, In, Zn, S, Si and O elements are clearly observed in TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> sample.



**Figure 1** Surface SEM images of the (a) CuInS<sub>2</sub>, (b) CuInS<sub>2</sub>/ZnS, and (c) CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> layers on the surface of TiO<sub>2</sub>.



**Figure 2** Elemental mapping images of the Cu, In, Zn, S, Si and O for the TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> samples.

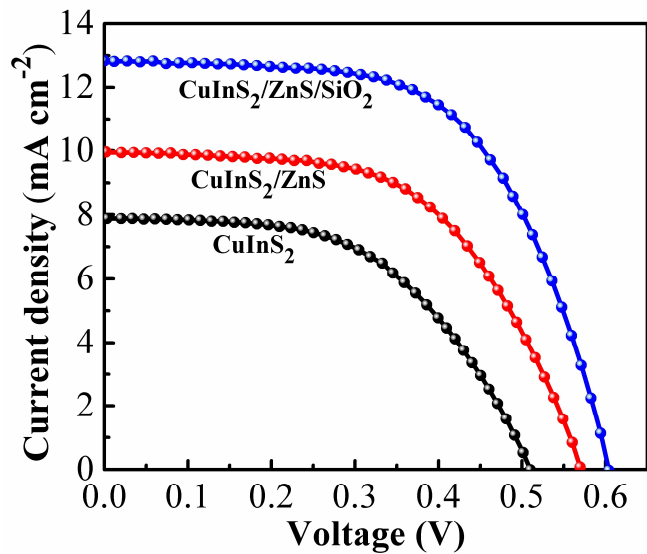


**Figure 3** (a) XPS survey of CuInS<sub>2</sub> QDs adsorbed on the TiO<sub>2</sub> film. XPS spectrum of close-up survey in the (b) Cu 2p region, (c) In 3d region, (d) Zn 2p region, (e) S 2p region, and (f) Si 2p region.

The composition of the CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> sample was investigated by XPS, as shown in Figure 3. The XPS survey spectra in Figure 3a depict peaks for Ti<sub>2p</sub>, O<sub>1s</sub>, C<sub>1s</sub>, Cu<sub>2p</sub>, In<sub>3d</sub>, Zn<sub>2p</sub>, S<sub>2p</sub>, and Si<sub>2p</sub>, respectively. The binding energy of Cu 2p<sub>3/2</sub> and Cu 2p<sub>1/2</sub> were observed at 932.9 and 952.7 eV, respectively (Figure 3b), with no evident shake-up satellite signals in this Cu<sub>2p</sub> spectrum. The two main peaks of In are located at 445.4 eV and 452.9 eV for In 3d<sub>5/2</sub> and In 3d<sub>3/2</sub>, respectively (Figure 3c). Figure 3d depicts the binding energies for Zn 2p<sub>3/2</sub> and Zn 2p<sub>1/2</sub> of the prepared sample of CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> at 1023.2 eV and 1046.2 eV respectively. The spectrum for S 2p was yielded peaks of S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub> at binding energies of 162.0 eV and 163.0 eV, respectively (Figure 3e). The Si 2p spectrum (Figure 3f) exhibits a main peak and shoulder at lower binding energies. The main peak observed at 103.2 eV has been ascribed to Si in the oxidized form (SiO<sub>2</sub>) and the other shoulder peak can be ascribed to the presence of crystalline Si (elemental Si).



2.2. Electrochemical characterization

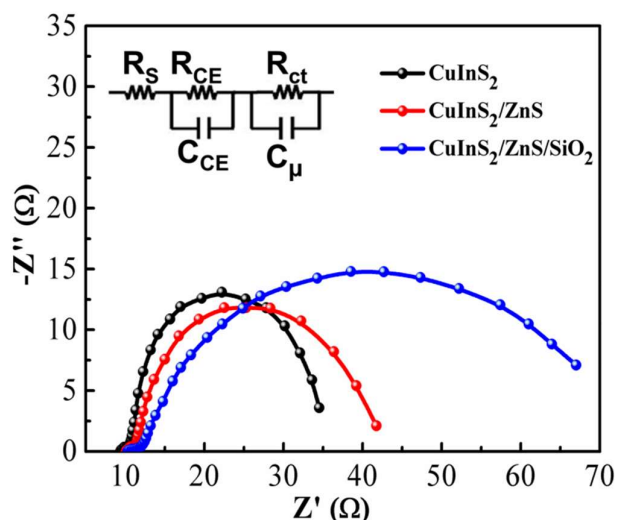


**Figure 4** Current density-voltage (J-V) characteristics of CuInS<sub>2</sub>, CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> QDSSCs measured under the illumination of one sun (AM 1.5, 100 mW cm<sup>-2</sup>).

The current density-voltage (J-V) curves of the QDSSCs based on various photoanodes were measured under AM 1.5 illumination (100 mW cm<sup>-2</sup>) are shown in Figure 4 and the corresponding photovoltaic parameters are tabulated in Table 1. When only CuInS<sub>2</sub> are deposited on TiO<sub>2</sub> film, the QDSSC exhibits a J<sub>sc</sub> of 7.87 mA cm<sup>-2</sup>, V<sub>oc</sub> of 0.509 V, FF of 0.537, resulting a low PCE of 2.15%. However, when ZnS and ZnS/SiO<sub>2</sub> passivation layers were deposited, all the photovoltaic parameters were greatly improved; the QDSSCs with a ZnS/SiO<sub>2</sub> layer exhibit the good performance, with J<sub>sc</sub>, V<sub>oc</sub>, and FF reaching 12.83 mA cm<sup>-2</sup>, 0.603 V, and 0.598, respectively and the highest PCE of 4.63%, which is much higher than the PCE of 3.23% with a ZnS passivation layer. It is observed that the performance of QDSSC with a double layer of ZnS/SiO<sub>2</sub> is higher than that of ZnS passivation layer, which is due to suppression of recombination losses in QDSSCs and increases the charge collection efficiency.

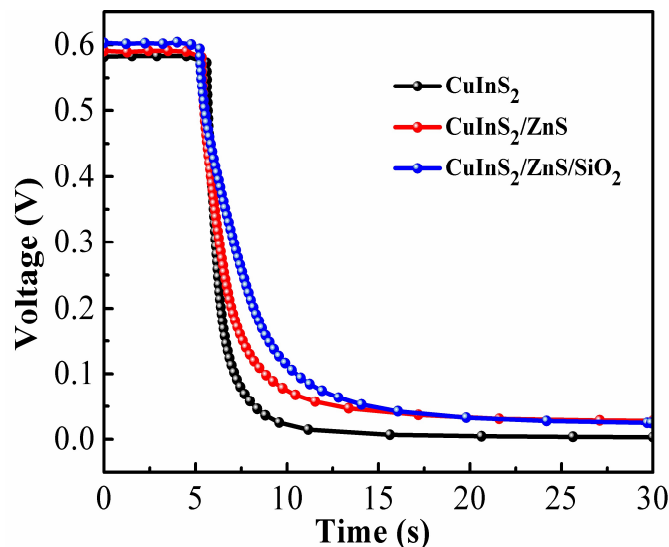
**Table 1** Summary of the photovoltaic properties and EIS results of QDSSCs fabricated various sensitized conditions

Cell	V <sub>oc</sub> (V)	J <sub>sc</sub> (mA cm <sup>-2</sup> )	FF	η%	R <sub>s</sub> (Ω)	R <sub>CE</sub> (Ω)	R <sub>ct</sub> (Ω)
CuInS <sub>2</sub>	0.509	7.87	0.537	2.15	9.34	0.84	30.65
CuInS <sub>2</sub> /ZnS	0.569	9.95	0.571	3.23	10.03	1.03	23.95
CuInS <sub>2</sub> /ZnS/SiO <sub>2</sub>	0.603	12.83	0.598	4.63	10.24	1.16	55.02



**Figure 5** EIS curves of QDSSCs based CuInS<sub>2</sub> and CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> cells in the form of Nyquist-plots and the inset shows the equivalent circuit used to fit the impedance spectra.

Electrochemical impedance spectroscopy (EIS) characterizations were conducted to investigate the charge recombination processes in devices under forward bias ( $V_{oc}$ ) and dark condition. Figure 5 depicts the EIS spectra of CuInS<sub>2</sub> QDSSCs without and with ZnS or ZnS/SiO<sub>2</sub> passivation layers which were fitted using Z-view software with the equivalent circuit provided in the inset of Figure 5. The corresponding fitting results are shown in Table 1. The Nyquist plot consists of two semicircles and the first semicircle represents the resistance at the CE/electrolyte interface. The second semicircle represents the charge transfer resistance at the interface of the TiO<sub>2</sub>/QDs/electrolyte. The intercept on the real axis of the impedance plot at higher frequency corresponds to the series resistance ( $R_s$ ) of FTO substrate and the resistance of FTO/TiO<sub>2</sub> [19,20]. It is noticed that there are no apparent differences observed in the  $R_s$  and  $R_{CE}$  due to the same CE and electrolyte used in these experiments. However, there is a noticeable difference in  $R_{ct}$ ; the  $R_{ct}$  value for the CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> based QDSSCs is 55.02  $\Omega$ , while  $R_{ct}$  value for the CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub> based QDSSCs are only 30.65  $\Omega$  and 23.95  $\Omega$ , respectively. The charge recombination resistance at the interface of TiO<sub>2</sub>/QDs/electrolyte is mainly determined by  $R_{ct}$ . The higher  $R_{ct}$  value represents the reduced recombination of the electrons and holes, and enhances the electron transfer process at TiO<sub>2</sub>/QDs/electrolyte interface. Therefore, it is confirmed that the deposition of ZnS/SiO<sub>2</sub> passivation layer on the CuInS<sub>2</sub> QDs favors the efficient electron transfer from CuInS<sub>2</sub> to TiO<sub>2</sub> photoanodes with suppression of the interfacial charge recombination processes, which is more effectively than that of the ZnS passivation layer.

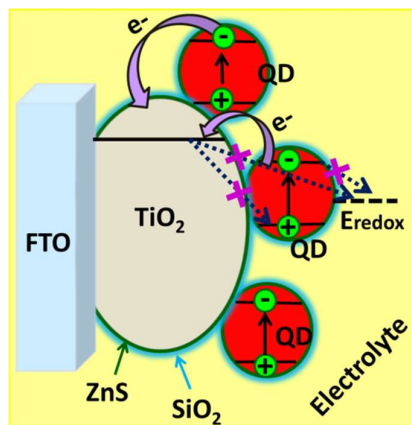


**Figure 6** Open circuit decay curves of QDSSCs based CuInS<sub>2</sub> and CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> cells.

Furthermore, open circuit voltage decay (OCVD) measurements were carried out to investigate the charge recombination process in QDSSCs and the results. OCVD analysis of QDSSCs was performed during relaxation from an illuminated quasiequilibrium state to the darkness. Figure 6 depicts the OCVD plots of the QDSSCs based on CuInS<sub>2</sub>, CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> photoanodes. Apparently, the CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> cell delivered considerably longer decay times than the CuInS<sub>2</sub> and CuInS<sub>2</sub>/ZnS cells, indicating a suppression of charge recombination process. Moreover, the  $V_{oc}$  decay is directly related to the electron life time according to following equation [21]:

$$\tau_e = -\left(\frac{k_B T}{e}\right) \left(\frac{dV_{oc}}{dt}\right)^{-1} \quad (1)$$

where  $k_B$  is the Boltzmann constant,  $T$  is the absolute temperature, and  $e$  is the electronic charge. Figure 5b depicts the electron life time ( $\tau_e$ , in a log representation) as a function of  $V_{oc}$ . It can be noticed that the  $\tau_e$  of all the devices increases with decreasing  $V_{oc}$ . Among the QDSSCs investigated, the CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> delivers longer  $\tau_e$  than the CuInS<sub>2</sub>/ZnS and CuInS<sub>2</sub> devices, implying suppressed recombination and efficient electron transfer at the TiO<sub>2</sub>/QDs/electrolyte, which is consistent with EIS analysis. Therefore, the slower  $V_{oc}$  decay and longer  $\tau_e$  of the CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> device efficiently suppressed the electron recombination from TiO<sub>2</sub> and QDs to electrolyte and higher charge collection efficiency contribute to the enhanced photovoltaic performance.



**Figure 7** Schematic illustration of the possible charge transfer mechanism in the  $\text{TiO}_2/\text{CuInS}_2/\text{ZnS}/\text{SiO}_2$  QDSSCs.

Several paths for charge recombination occurs in the QDSSCs, in which the excited electrons from the  $\text{TiO}_2$  can react with the electrolyte or with the QDs and photo excited electrons from the QDs react with the electrolyte. The deposition of  $\text{ZnS}/\text{SiO}_2$  layer on the surface of  $\text{TiO}_2/\text{CuInS}_2$  can effectively suppress the charge recombination process at the photoanode/electrolyte interface (Figure 7) and improve the photovoltaic performance of QDSSCs.

#### 4. Materials and Methods

##### 4.1. Preparation of $\text{TiO}_2$ electrodes

$\text{TiO}_2$  film was fabricated by doctor blading the  $\text{TiO}_2$  paste (20 nm, Ti-Nanoxide HT/TP, Solaronix) on well-cleaned fluorine-doped tin oxide (FTO,  $1.3 \times 1.6 \text{ cm}^2$ ) substrate with an active area of  $0.25 \text{ cm}^2$ . The film was then heated at  $450^\circ\text{C}$  for 30 min and producing a thickness of  $7.5 \mu\text{m}$  after solvent evaporation [22].

##### 4.1. Deposition of $\text{CuInS}_2$ QDs on $\text{TiO}_2$ electrodes

$\text{CuInS}_2$  QDs was deposited on the  $\text{TiO}_2$  substrate using the successive ion layer adsorption and reaction (SILAR) process.  $\text{TiO}_2$  electrodes were successively immersed into the three different solutions: one of 0.05 M of  $\text{Cu}(\text{NO}_3)_2$  for 2 min, another of 0.05 M of  $\text{In}(\text{NO}_3)_3$  for 2 min, and a final one of 0.1 M of  $\text{Na}_2\text{S}$  for 5 min. Following each immersion, the films were rinsed with deionized (DI) water for 1 min to remove the excess precursors. This procedure comprises one  $\text{CuInS}_2$  SILAR cycle and was repeated six times.

##### 4.3. Deposition of $\text{ZnS}$ , $\text{ZnS}/\text{SiO}_2$ passivation layers on $\text{TiO}_2/\text{CuInS}_2$ electrodes

The  $\text{ZnS}$  passivation layer was deposited on  $\text{TiO}_2/\text{CuInS}_2$  electrodes by a SILAR process. Typically,  $\text{TiO}_2/\text{CuInS}_2$  electrodes were successively immersed into aqueous solutions containing 0.1 M of  $\text{Zn}(\text{NO}_3)_2$  and 0.1 M  $\text{Na}_2\text{S}$  for 1 min, respectively. This process was repeated for three times and the electrode is named as  $\text{CuInS}_2/\text{ZnS}$ . Furthermore,  $\text{SiO}_2$  coating was carried out by dipping the



CuInS<sub>2</sub>/ZnS electrodes in 0.01 M tetraethylorthosilicate ethanol solution containing 0.1 M NH<sub>4</sub>OH for 1 h. The as-fabricated electrode is termed as CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub>.

#### 4.4. QDSSC fabrication

CuS counter electrode (CE) on FTO substrate was prepared according to the literature [23]. The photoanode was clamped with CuS CE using sealant ((SX 1170-60, Solaronix) at 100 °C) and the space between the electrodes was filled with polysulfide electrolyte (1 M Na<sub>2</sub>S, 2 M S, and 0.2 M KCl in methanol and water at a ratio of 7:3).

#### 4.5. Characterizations

The surface morphology and elemental composition of the electrodes were investigated using a scanning electron microscopy (SEM, S-2400, Hitachi) equipped with energy-dispersive X-ray spectroscopy (EDX) operated at 15 kV. The current-voltage (J-V) characteristics of the QDSSCs were examined under one sun illumination (AM 1.5G, 100 mW cm<sup>-2</sup>) using an ABET Technologies (USA) solar simulator with an irradiance uniformity of ±3%. Electrochemical impedance spectroscopy (EIS) was investigated using a BioLogic potentiostat/galvanostat/EIS analyzer (SP-150, France) under one sun illumination with a frequency of 100 mHz-500 kHz and 10 mV of AC amplitude.

### 5. Conclusions

Introduction of ZnS/SiO<sub>2</sub> passivation layer on TiO<sub>2</sub>/CuInS<sub>2</sub> QDs has been demonstrated to be an efficient and promising approach to significantly suppress the charge recombination at photoanode/electrolyte and enhance the power conversion efficiency. Eventually, an overall  $\eta$  of 4.63% was obtained for the TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS/SiO<sub>2</sub> QDSSCs, which is approximately 43% enhancement over the  $\eta$  = 3.23% for the TiO<sub>2</sub>/CuInS<sub>2</sub>/ZnS QDSSCs and more than 115% increment over the  $\eta$  = 2.15% for the device without a passivation layer (TiO<sub>2</sub>/CuInS<sub>2</sub>). Overall, the method of a ZnS/SiO<sub>2</sub> passivation layer is an effective approach to enhance the overall power conversion efficiency of QDSSCs.

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**Conflicts of Interest:** The authors declare no conflict of interest.

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